Octaèdres NbO ₆			
Nb—O(3) $\times 2$	1,928 (4)	$O(3) \cdot \cdot \cdot O(2) \times 2$	2,79 (1)
Nb-0(4) × 2	2,000 (9)	$O(3) \cdot \cdot \cdot O(2) \times 2$	2,86 (1)
Nb-O(2) $\times 2$	2,0099 (8)	O(4)· · ·O(4)	2,77 (2)
O(3)···O(3)	2,78 (3)	$O(4) \cdot \cdot \cdot O(2) \times 2$	2,85 (1)
$O(3) \cdot \cdot \cdot O(4) \times 2$	2,79 (2)	$O(4) \cdot \cdot \cdot O(2) \times 2$	2,73 (2)
$O(3) \cdot \cdot \cdot O(4) \times 2$	3,92 (1)	O(2)· · ·O(2)	4,016 (2)
O(3)-Nb-O(3)	92,1 (8)	O(4)—Nb—O(4)	87,5 (7)
$O(4)$ —Nb— $O(3) \times 2$	90,3 (5)	$O(2)$ —Nb— $O(4) \times 2$	90,7 (5)
$O(4)$ —Nb— $O(3) \times 2$	175,4 (4)	$O(2)$ —Nb— $O(4) \times 2$	85,7 (5)
$O(2)$ —Nb— $O(3) \times 2$	90,2 (5)	O(2)—Nb—O(2)	175,1 (9)
$O(2)$ —Nb— $O(3) \times 2$	93,2 (4)		
Polyèdres KO ₁₂			
$K \cdot \cdot \cdot O(4) \times 2$	2,84 (1)	$\mathbf{K} \cdot \cdot \cdot \mathbf{O}(2)$	3,19 (2)
$K \cdot \cdot \cdot O(3) \times 2$	2,85 (1)	K· · · O(2)	3,24 (2)
$K \cdot \cdot \cdot O(4) \times 2$	2,93 (1)	$K \cdot \cdot \cdot O(4) \times 2$	3,25 (1)
K···O(1)	2,990 (5)	$K \cdot \cdot \cdot O(4) \times 2$	3,48 (1)
$K \cdot \cdot \cdot O(2)$	3,08 (2)		

Programme(s) pour la solution de la structure: *MULTAN80* (Main *et al.*, 1980). Programme(s) pour l'affinement de la structure: *ORXFLS* (Busing, 1971) modifié par P. Laruelle. Les dessins ont été obtenus à l'aide de *MOLVIEW* (Cense, 1990). Programme(s) pour la préparation du manuscrit: *ORFFE* (Busing, Martin & Levy, 1964).

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Les listes des facteurs de structure et des facteurs d'agitation thermique anisotrope ont été déposées au dépôt d'archives de l'UICr (Référence: DU1095). On peut en obtenir des copies en s'adressant à: The Managing Editor, International Union of Crystallography, 5 Abbey Square, Chester CH1 2HU, Angleterre.

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$CaV_2O(PO_4)_2$, Isotypic with the Cd Phase

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Abstract

The new V^{III} monophosphate, calcium divanadium oxodiphosphate, $CaV_2O(PO_4)_2$, is isotypic with the Cd phase. Its structure consists of $[V_2PO_7]_{\infty}$ layers, built up from $[VO_4]_{\infty}$ rutile chains and $[VPO_8]_{\infty}$ chains, interconnected through PO₄ tetrahedra. This framework delimits tunnels running along $\langle 011 \rangle$, at the intersection of which the calcium ions are located in a '6 + 1' coordination mode.

Comment

Investigations of vanadium(III) phosphates performed in the last ten years have shown that it is possible to generate original mixed frameworks with $[VO_4]_{\infty}$ rutile chains. Nevertheless, the formation of such infinite chains remains rare since only two examples. VPO₄ (Kinomura, Muto & Koizumi, 1982) and CdV₂O(PO₄)₂ (Boudin, Grandin, Borel, Leclaire & Raveau, 1994), are known as yet. Calcium, owing to the similarity of its crystal chemistry with that of cadmium, should be considered for the generation of such structures. Two compounds, CaV₂P₂O₁₀ (Lii, Chen, Kang & Wang, 1992) and CaV₂(P₂O₇)₂ (Hwu & Willis, 1991), which are isotypic with the Cd phases $CdV_2P_2O_{10}$ (Leclaire, Chardon, Grandin, Borel & Raveau, 1993) and $CdV_2(P_2O_7)_2$ (Boudin, Grandin, Leclaire, Borel & Raveau, 1994), respectively, have already been synthesized, whereas Ca₂V(PO₄)(P₂O₇) (Lii, Wen, Su & Chueh, 1992) does not exhibit structural relationships with the Cd phase. We have investigated the system Ca-V^{III}-P-O, and we report here the synthesis and crystal structure of a new V^{III} monophosphate, $CaV_2O(PO_4)_2$, isotypic with the cadmium phase.

The results (Table 1) clearly show that $CaV_2O(PO_4)_2$ is isotypic with $CdV_2O(PO_4)_2$. Thus, its structure is built up from $[V_2P(1)O_7]_{\infty}$ infinite waving layers parallel to (100) in which one $[V(2)O_4]_{\infty}$ rutile chain of edgesharing octahedra running along b alternates with two $[V(1)PO_8]_{\infty}$ chains of corner-sharing $V(1)O_6$ octahedra and $P(1)O_4$ tetrahedra (Fig. 1). Along **a**, the $[V_2PO_7]_{\infty}$ layers are connected through $P(2)O_4$ tetrahedra in such

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Fig. 1. Projection of the structure of $CaV_2O(PO_4)_2$ along **b** showing the stacking of the $[V_2PO_7]_{\infty}$ and $[PO_2]_{\infty}$ layers.

a way that one $P(2)O_4$ tetrahedron shares one edge with one $V(2)O_6$ octahedron of one layer and two apices with two $V(2)O_6$ octahedra of the next layer.

In this framework, as in the cadmium vanadium phosphate, the VO₆ octahedra are not strongly distorted; this can be seen for the V(1) octahedra which form the $[VPO_8]_{\infty}$ chains and exhibit V(1)—O distances ranging from 1.898 (4) to 2.164 (4) Å (Table 2) compared with 1.90-2.17 Å for the Cd phase (Boudin, Grandin, Borel, Leclaire & Raveau, 1994). Note, however, that the V(2) octahedra of the rutile chains, in spite of their identical ranges of bond lengths in the Ca and Cd phases [2.008(2)-2.071(3)] Å in the Ca phase and 1.98-2.07 Å in Cd], show different behaviour. In the Ca compound, one observes four short bonds (average 2.011 Å) and two longer ones [2.071 (3) Å], while in the Cd compound one observes two short bonds (1.98 Å) and four longer ones (average 2.07 Å). The geometries of the PO₄ tetrahedra are very similar in both structures, *i.e.* the P(1) tetrahedra of the $[VPO_8]_{\infty}$ chains are very regular with four P-O distances close to 1.54 Å, whereas the P(2) tetrahedra that share one edge with VO₆ octahedra exhibit three longer P-O distances (1.53-1.54 Å) and a shorter one of 1.50 Å.

Like cadmium, calcium is located at the intersection of small tunnels running along the (011) directions. This cation, like cadmium, exhibits distorted octahedral coordination, with Ca-O distances ranging from 2.342(3) to 2.519(5) Å (Table 2), which are greater than

the Cd-O distances (2.28-2.46 Å), consistent with the larger size of calcium. Note, however, that calcium has a seventh neighbour at 2.911 (4) Å, whereas for cadmium the seventh oxygen neighbour is much further away (more than 3 Å). Taking into consideration this seventh neighbour leads to an electrostatic valence of 1.92 using the Brown & Altermatt (1985) curves, instead of 1.84 with six oxygen neighbours. This difference between calcium and cadmium is easily understood on the basis of the larger ionicity of calcium with respect to cadmium. In conclusion, $CaV_2O(PO_4)_2$ represents the third example presently known of a V^{III} monophosphate with rutile chains.

Experimental

A single crystal of CaV₂O(PO₄)₂ was extracted from a mixture of composition 'CaV₂P₂O₈'. The latter was prepared by first heating CaCO₃, V₂O₅ and H(NH₄)₂PO₄ at 1073 K in air for 2h and then adding to the intermediate mixture the appropriate amount of metallic V. The final mixture was placed in an alumina crucible, heated in an evacuated silica ampoule at 1453K for 24h, slowly cooled at a rate of 4 K h^{-1} to 1253 K and finally quenched to room temperature. The composition of the black lamellar single crystal was confirmed by microprobe analysis.

Crystal data

Mo $K\alpha$ radiation
$\lambda = 0.71073 \text{ Å}$
Cell parameters from 25
reflections
$\theta = 18-22^{\circ}$
$\mu = 3.968 \text{ mm}^{-1}$
T = 294 K
Lamellar
0.077 \times 0.051 \times 0.039 mm
Black

 $\theta_{\rm max} = 45^{\circ}$

 $h = 0 \rightarrow 28$

 $k = 0 \rightarrow 12$

 $l = 0 \rightarrow 14$

3 standard reflections

frequency: 50 min

intensity decay: none

Data collection Enraf-Nonius CAD-4 diffractometer Bisect scans Absorption correction: none 3133 measured reflections 3133 independent reflections 914 observed reflections $[I > 3\sigma(I)]$

Refinement

Refinement on F	$(\Delta/\sigma)_{\rm max} < 0.004$
R = 0.030	$\Delta \rho_{\rm max} = 0.841 \text{ e } \text{\AA}^{-3}$
wR = 0.032	$\Delta \rho_{\rm min} = -1.251 \ {\rm e} \ {\rm \AA}^{-3}$
S = 0.999	Atomic scattering factors
914 reflections	from International Tables
79 parameters	for X-ray Crystallography
$w = F\sin(\theta/\lambda)$	(1974, Vol. IV)

Table	1. Frac	tional	atomic	coor	dinates	and	equivalent
	isotro	pic dis	placem	ent p	aramete	rs (Å	²)

$B_{\rm eq} = (4/3)$	3) $\sum_i \sum_i \beta_{ii} \mathbf{a}_i \cdot \mathbf{a}_i$.
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	x	у	Z	Bea
Ca	0.21979 (7)	1/4	0.1149 (2)	0.70(1)
V(1)	0.37954 (5)	3/4	0.1489(1)	0.321 (9)
V(2)	0	1/2	0	0.404 (9)
P(1)	0.18974 (8)	3/4	0.1134 (2)	0.46 (2)
P(2)	0.43270 (8)	1/4	0.1366 (2)	0.36 (2)
O(1)	0.3662 (2)	0.4378 (4)	0.1278 (4)	0.65 (3)
O(2)	0.2632 (3)	3/4	-0.0424 (6)	0.77 (6)
O(3)	0.4929 (3)	3/4	0.0081 (7)	1.23 (6)
O(4)	0.2514 (3)	3/4	0.2875 (5)	0.59 (5)
O(5)	0.4442 (2)	3/4	0.3759 (5)	0.59 (5)
O(6)	-0.0186 (3)	1/4	0.1750 (5)	0.77 (6)
O(7)	0.1283 (2)	0.5533 (4)	0.1084 (4)	0.79 (4)

Table 2. Selected geometric parameters (Å, °)

Ca = O(7)	2.342 (3)	$O(5^n) \cdot \cdot \cdot O(6)$	2.430 (5)
$Ca - O(7^i)$	2.342 (3)	O(5 ⁱⁱ)· · ·O(6 ^{vi})	3.276(1)
Ca-0(1)	2,405 (3)	O(5 ⁱⁱ) · · ·O(7 ^{vi})	2.905 (4)
C_{a}	2.405 (3)	$O(5^{ii}) \cdots O(7)$	2 784 (4)
C_{a} $O(4^{ii})$	2 430 (4)	V(2) = O(6)	2 071 (3)
$C = O(2^{11})$	2 519 (5)	$O(6) \cup O(6^{3})$	4 142 (7)
$C_{a} = O(2^{i})$	2.517(3)		4.142 (7)
	2.911 (4)	$O(0) \cdots O(7^n)$	2.884 (5)
v(1) = O(1)	2.020 (3)	$O(6) \cdots O(7)$	2.895 (4)
$O(1) \cdots O(1^{*})$	4.011 (7)	$V(2) = O(6^{v_1})$	2.071 (3)
O(1)···O(2)	2.777 (4)	$O(6^{v_1}) \cdots O(7^{v_1})$	2.895 (4)
O(1)···O(3)	2.832 (4)	$O(6^{v_1}) \cdot \cdot \cdot O(7)$	2.884 (5)
O(1)···O(4)	2.836 (4)	V(2)—O(7 ^{vi})	2.015 (3)
O(1)· · ·O(5)	2.922 (4)	$O(7^{v_1}) \cdots O(7)$	4.031 (5)
$V(1) = O(1^{iv})$	2.020 (3)	V(2) - O(7)	2.015 (3)
$O(1^{iv}) \cdots O(2)$	2.777 (4)	P(1) - O(2)	1 545 (4)
$O(1^{iv}) \cdots O(3)$	2 832 (4)	$\Omega(2) \cdots \Omega(4)$	2 420 (6)
$O(1^{iv}) + O(4)$	2.836 (4)	$O(2) \cup O(7)$	2.420(0)
$O(1^{iv})$, $O(5)$	2,000 (1)	$O(2) \dots O(7^{N})$	2.545(4)
V(1) = O(2)	2.322(4)	P(1) = O(4)	2.343 (4)
V(1) = O(2)	2.104 (4)	P(1) = O(4)	1.545 (4)
$O(2) \cdots O(3)$	3.281 (0)	$O(4) \cdots O(7)$	2.523 (4)
$O(2) \cdots O(4)$	2.420 (0)	$O(4) \cdots O(7^{n})$	2.523 (4)
$U(2) \cdots U(5)$	3.995 (7)	P(1) = O(7)	1.536(3)
V(1) = O(3)	1.911 (5)	$O(7) \cdots O(7^{n})$	2.527 (6)
$O(3) \cdots O(4)$	3.992 (7)	$P(1) = O(7^{*})$	1.536 (3)
$O(3) \cdots O(5)$	2.779 (7)	P(2) - O(1)	1.533 (3)
V(1)—O(4)	2.083 (4)	$O(1) \cdot \cdot \cdot O(1^{\circ})$	2.413 (5)
O(4)· · ·O(5)	2.811 (5)	$O(1) \cdot \cdot \cdot O(3^{vu})$	2.537 (5)
V(1)-O(5)	1.898 (4)	O(1)· · · O(6 ^{viii})	2.491 (4)
V(2)—O(5 ^v)	2.008 (2)	$P(2) - O(1^{i})$	1.533 (3)
O(5 ^v)· · · O(5 ⁱⁱ)	4.016 (5)	O(1 ⁱ)· · · O(3 ^{vii})	2.537 (5)
O(5 ^v)· · · O(6)	3.276(1)	O(1 ⁱ)· · · O(6 ^{viii})	2.491 (4)
$O(5^v) \cdot \cdot \cdot O(6^{v_i})$	2.430 (5)	P(2)-O(3 ^{vii})	1.495 (5)
$O(5^v) \cdot \cdot \cdot O(7^{v_i})$	2.784 (4)	O(3 ^{vii})· · · O(6 ^{viii})	2.464 (7)
$O(5^v) \cdot \cdot \cdot O(7)$	2.905 (4)	$P(2) - O(6^{viii})$	1.542 (4)
$V(2) - O(5^{ii})$	2.008 (2)		
$U(1^{\circ}) = V(1) = U(1)$	100.2 (2)	$O(7^{(1)}) - V(2) - O(5^{(1)})$	87.6(1)
O(2) = V(1) = O(1)	83.1(1)	$O(7^{**}) - V(2) - O(5^{**})$	92.4 (1)
$O(2) = V(1) = O(1^{11})$	83.1 (1)	$O(7^{**}) - V(2) - O(6)$	89.8 (1)
O(3) - V(1) - O(1)	92.2 (1)	$O(7^{v_1}) - V(2) - O(6^{v_1})$	90.2 (1)
$O(3) - V(1) - O(1^{W})$	92.2 (1)	$O(7) - V(2) - O(5^{v})$	92.4 (1)
O(3)—V(1)—O(2)	107.1 (2)	$O(7) - V(2) - O(5^{u})$	87.6(1)
O(4)—V(1)—O(1)	87.5 (1)	O(7)—V(2)—O(6)	90.2(1)
$O(4) - V(1) - O(1^{iv})$	87.5 (1)	O(7)—V(2)—O(6 ^{vi})	89.8 (1)
O(4)—V(1)—O(2)	69.4 (2)	O(7)—V(2)—O(7 ^{vi})	180.0
O(4)—V(1)—O(3)	176.5 (2)	O(4) - P(1) - O(2)	103.1 (2)
O(5)-V(1)-O(1)	96.4 (1)	O(7) - P(1) - O(2)	111.5 (1)
$O(5) - V(1) - O(1^{iv})$	96.4 (1)	O(7) - P(1) - O(4)	109.9 (2)
O(5) - V(1) - O(2)	159.2 (2)	$O(7^{iv}) - P(1) - O(2)$	111.5 (1)
O(5) - V(1) - O(3)	93.7 (2)	$O(7^{iv}) \rightarrow P(1) \rightarrow O(4)$	109.9 (2)
O(5) - V(1) - O(4)	89.8 (2)	$O(7^{iv}) = P(1) = O(7)$	1108(2)
$O(5^{ii}) - V(2) - O(5^{v})$	180.0	$O(1^{i}) = P(2) = O(1)$	103 9 (2)
O(6) - V(2) - O(5'')	1069(1)	$O(3^{Vii}) = P(2) = O(1)$	1130(2)
$O(6) = V(2) = O(5^{ii})$	731(1)	$O(3^{vii}) = P(2) = O(1^{i})$	1130(2)
$O(6^{vi}) = V(2) = O(5^{vi})$	731(1)	$O(6^{VIII}) = P(2) = O(1)$	108 3 (2)
$O(6^{vi}) V(2) O(5^{ii})$	1060(1)	$O(6^{VIII}) = P(2) = O(1^{1})$	108.3 (2)
$O(6^{vi}) = V(2) = O(3^{vi})$	190.9 (1)	O(0) = P(2) = O(1)	100.5 (2)
U(0) = V(2) = U(0)	100.0	$U(0) = r(2) = U(3^{-1})$	108.3 (3)

Symmetry codes: (i) $x, \frac{1}{2} - y, z$; (ii) $\frac{1}{2} - x, y - \frac{1}{2}, z - \frac{1}{2}$; (iii) $\frac{1}{2} - x, y - \frac{1}{2}, \frac{1}{2} + z$; (iv) $x, \frac{3}{2} - y, z$; (v) $x - \frac{1}{2}, \frac{3}{2} - y, \frac{1}{2} - z$; (vi) -x, 1 - y, -z; (vii) 1 - x, 1 - y, -z; (viii) $\frac{1}{2} + x, \frac{1}{2} - y, \frac{1}{2} - z$.

Data were collected with $\omega - 4/3\theta$ scans of width $(1 + 0.35\tan\theta)^{\circ}$ and with and a slit aperture of $(1 + \tan\theta)$ mm. The refinements in $Pn2_1a$ (other setting of $Pna2_1$) failed. Calculations were performed using a MicroVAXII computer with the *SDP* system (B. A. Frenz & Associates, Inc., 1982).

Lists of structure factors and anisotropic displacement parameters have been deposited with the IUCr (Reference: DU1094). Copies may be obtained through The Managing Editor, International Union of Crystallography, 5 Abbey Square, Chester CH1 2HU, England.

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Hydroxylammonium Sulfate

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Abstract

The structure of $(NH_3OH)_2SO_4$ consists of sulfate and hydroxylammonium ions, which are connected by O—H···O and N—H···O hydrogen bonds. The bond distances fall within normal limits. All O atoms of the sulfate group are involved in two hydrogen bonds.

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